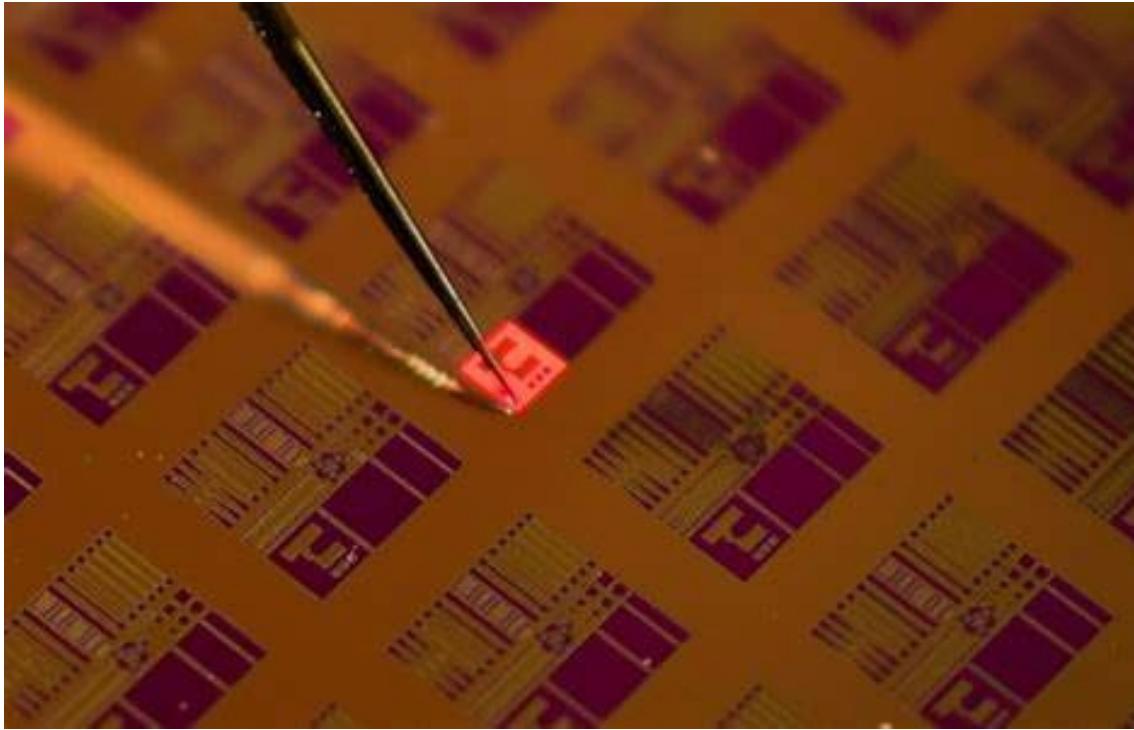

Advanced III-nitride semiconductor devices



<https://www.microled-info.com/micledi-demonstrates-a-lingap-ferrari-red-microled-devices>



LEDs

- Polarization field and LED efficiency
- Red microLEDs

LED efficiency

letters to nature

Acknowledgements

This work was supported in part by the National Science Foundation. We thank the Numerically Intensive Computing Group led by V. Agarwala, J. Holmes and J. Nucciarone, at the Penn State University CAC, for assistance and computing time with the LION-X cluster.

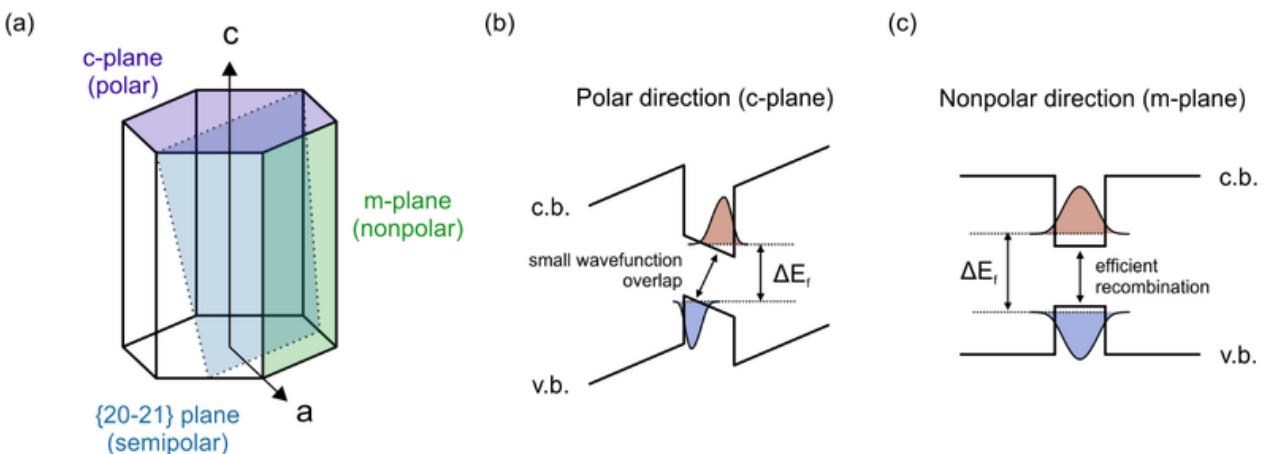
Correspondence and requests for materials should be addressed to J.K.J.
(e-mail: jain@phys.psu.edu).

Nitride semiconductors free of electrostatic fields for efficient white light-emitting diodes

**P. Walterelt, O. Brandt, A. Trampert, H. T. Grahn, J. Menniger,
M. Ramstelner, M. Reiche & K. H. Ploog**

*Paul-Drude-Institut für Festkörperelektronik, Hausvogteiplatz 5-7,
D-10117 Berlin, Germany*

The polarization field is commonly believed to be detrimental for the LED efficiency



Chapter Nonpolar and semipolar LEDs

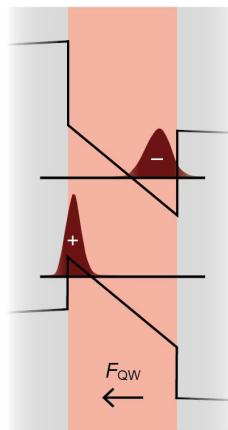
8.1.5 Advantages of nonpolar and semipolar LEDs

The QCSE imposes a limit on device performance for c-plane LEDs. To circumvent the detrimental effects of the internal polarization, growing devices along orientations that have zero or minimal polarization has been proposed as the solution for many unsolved issues. When there are no polarization-induced electric fields in the QW, the IQE is enhanced and there is an improved overlap between carrier wave functions. The QWs can be grown thicker without much reduction of the IQE as long as the defect density is not significantly increased. Therefore, the unbalanced carrier transport in MQW devices resulting from the over-thin QWs can be mitigated. Additionally, lower polarization

LED efficiency

Impact of the large internal electric field

$$F_{QW} = 2-3 \text{ MV/cm}$$

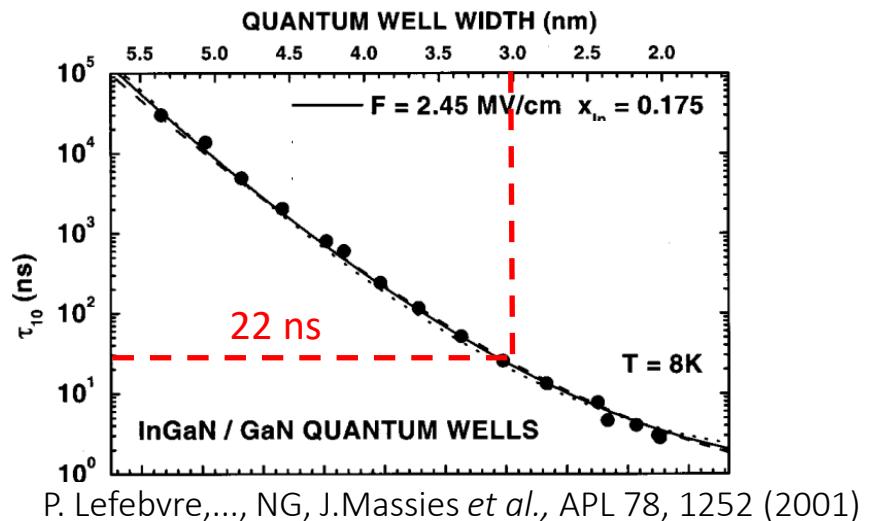


Quantum Confined Stark Effect (QCSE)

$$\text{Radiative lifetime: } \tau_R \propto \frac{1}{|\langle \varphi_e | \varphi_h \rangle|^2}$$

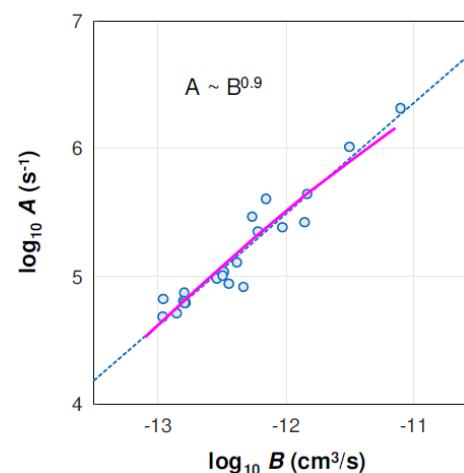
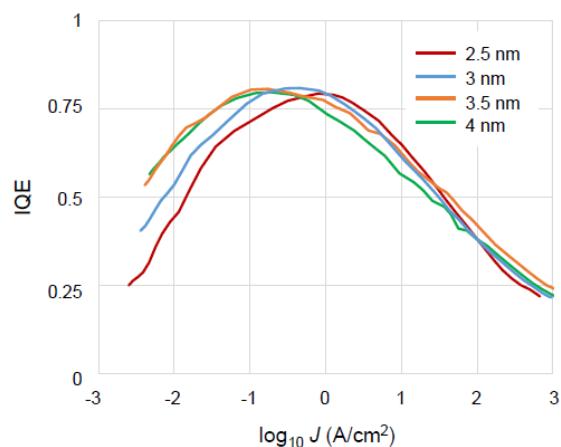
Few 10s of nanoseconds for blue LEDs

$$IQE = \frac{\tau_{NR}}{\tau_{NR} + \tau_R}$$



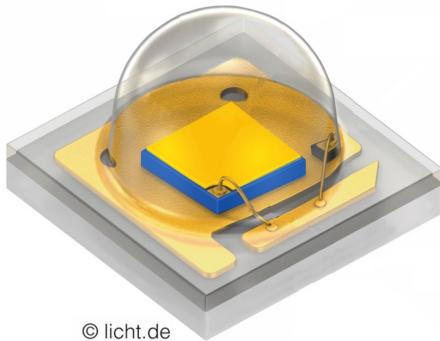
Important: τ_{NR} is inversely proportional to the e-h wavefunction overlap

A. David et al., Phys. Rev. Appl. 11, 031001 (2019)

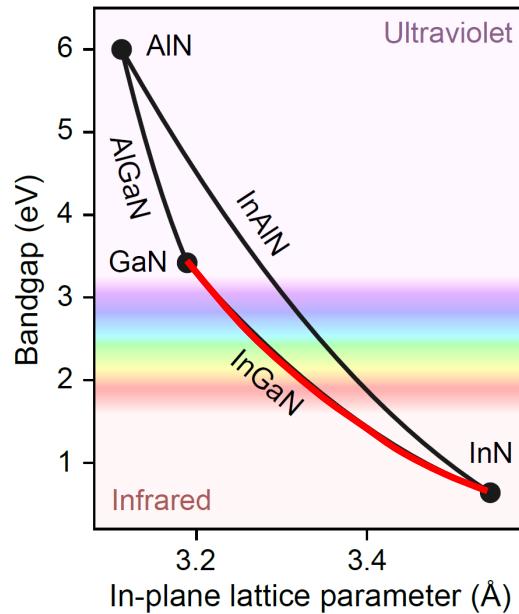


The B coefficient decreases when the polarization field increases, but the A coefficient decreases too.

The polarization field does not impact the efficiency.



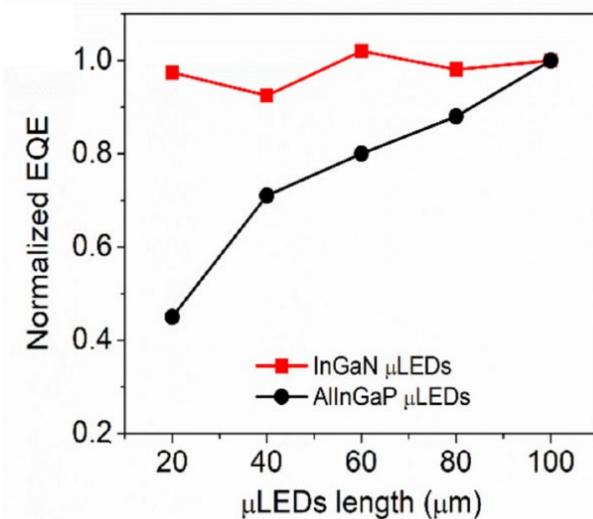
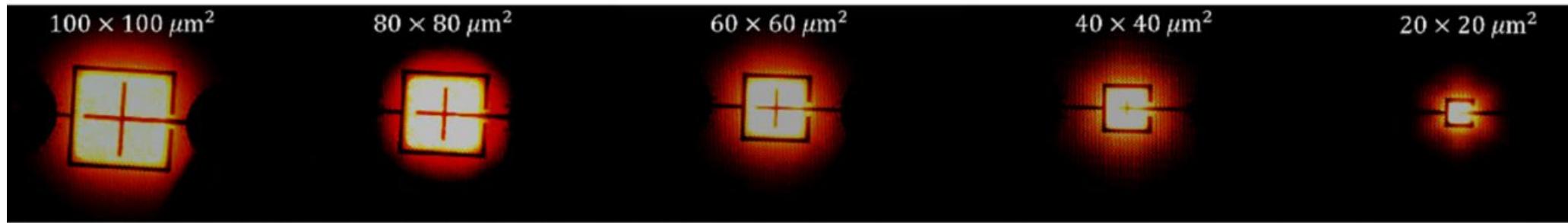
- Blue LEDs are extremely efficient
- WPE > 80% (IQE > 90%)
- White LEDs with record luminous efficacy of 300 lm/W



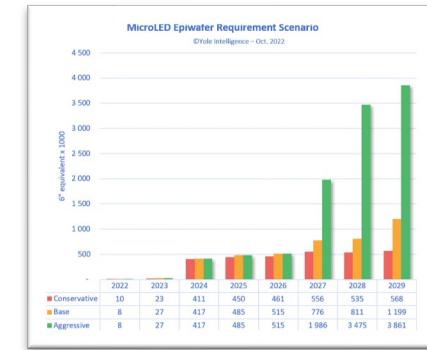
- InN bandgap of 0.65 eV
- ⇒ InGaN alloy covers the whole visible spectrum

- InGaN MicroLEDs for RGB displays

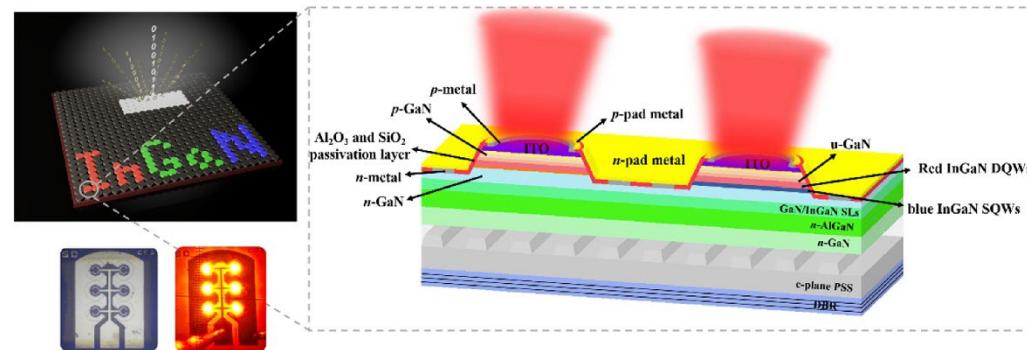
1. Less sensitive to sidewall non-radiative recombination



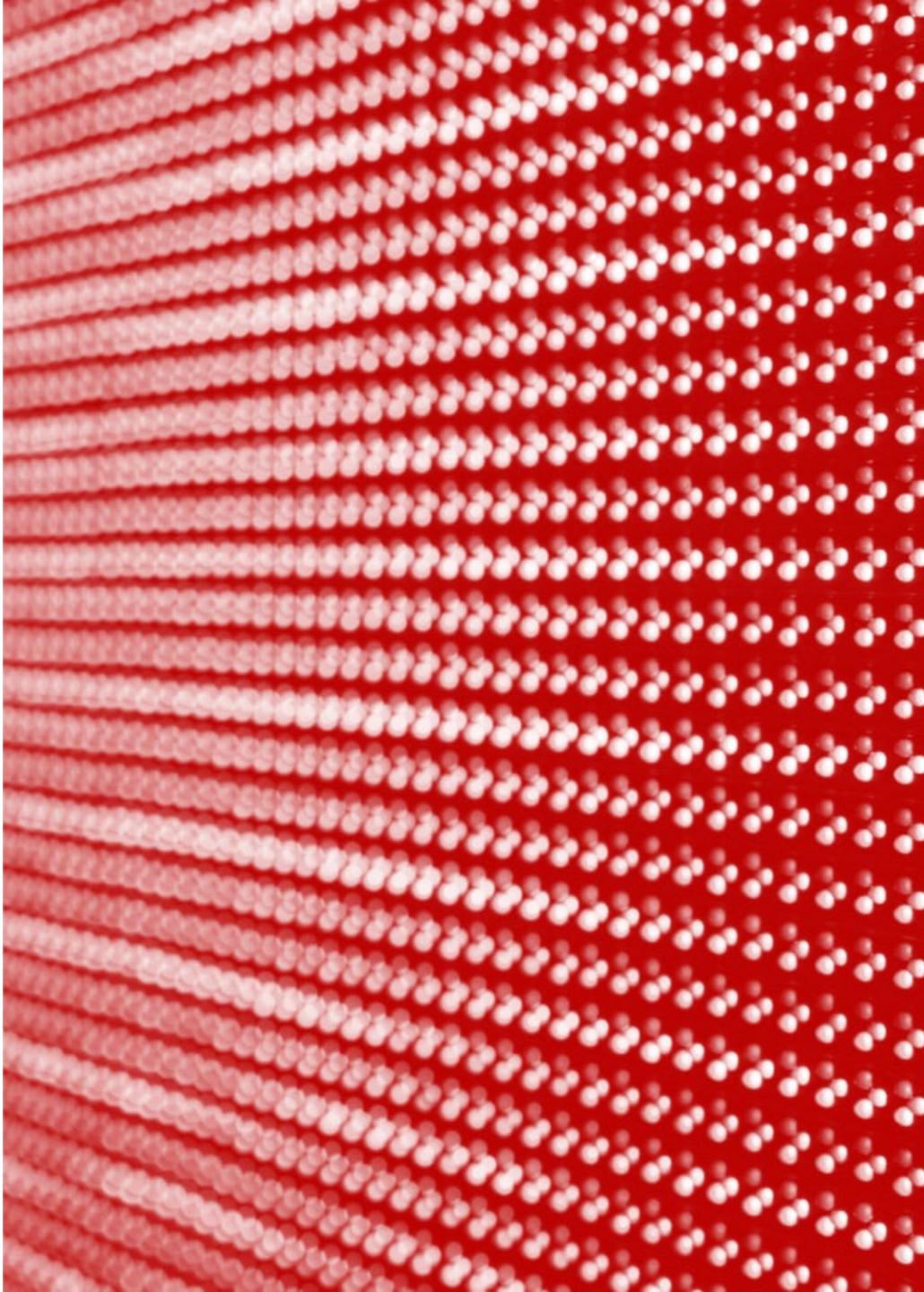
P. Li *et al.*, Crystals **12**, 541 (2022)



2. A single material platform



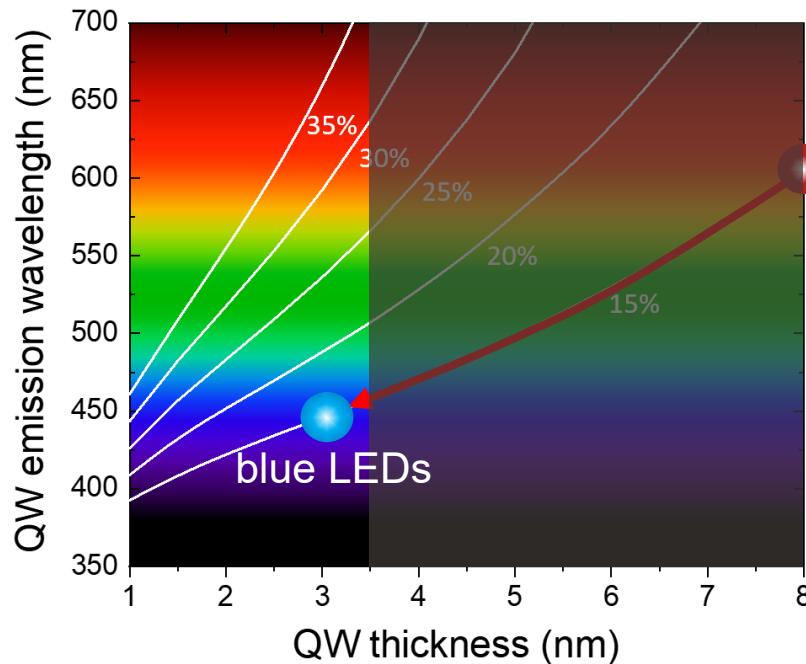
Y.-M. Huang *et al.*, Photonics Research **10**, 1978 (2022)



**How to push the
wavelength to the
Red ?**

How to push the wavelength to the Red ?

- InGaN/GaN QW emission wavelength

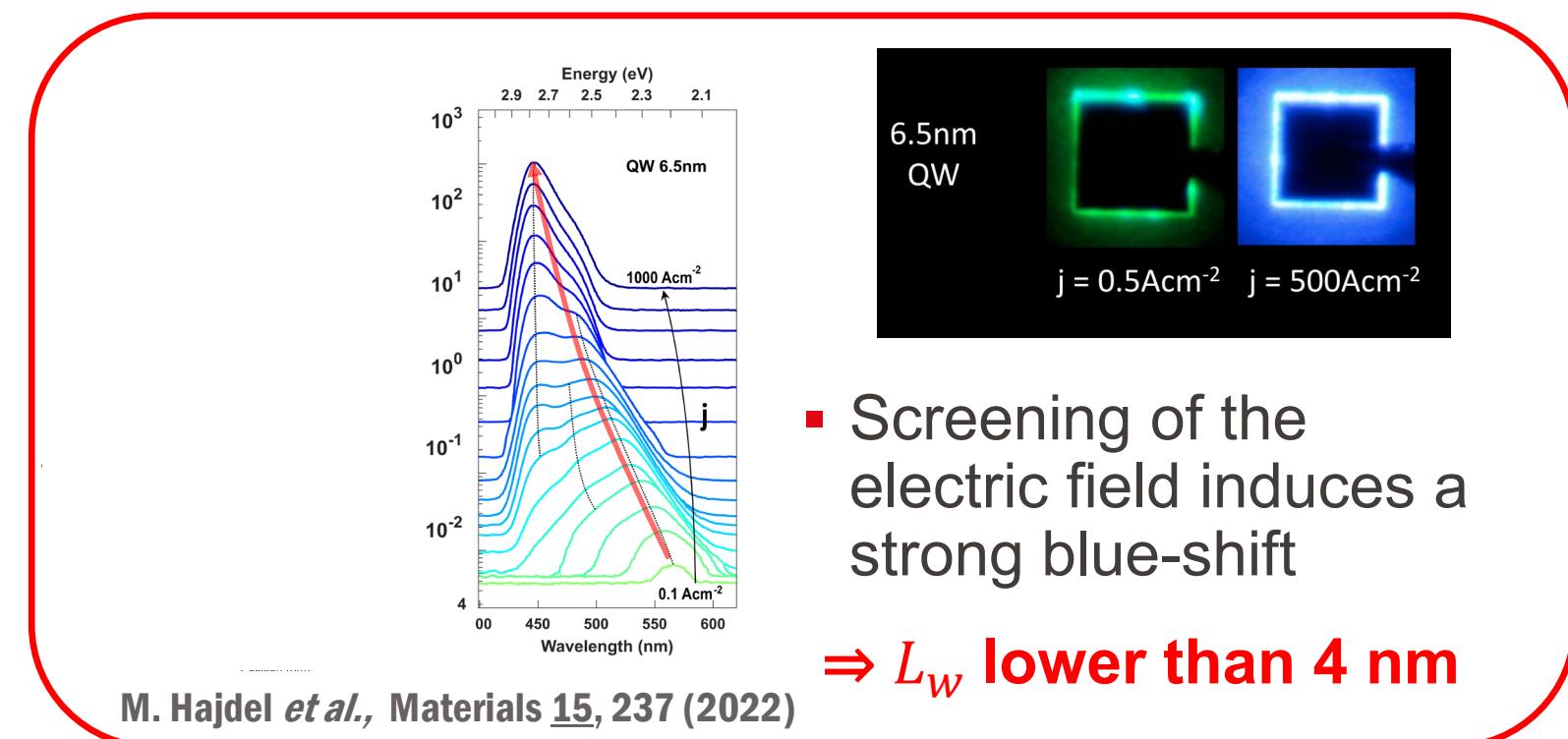


The indium content must be larger than 30%

$$E_{QW} = E_g^{InGaN} + E_{conf}^{e,h} \boxed{- qL_w F} \quad \text{Stark shift}$$

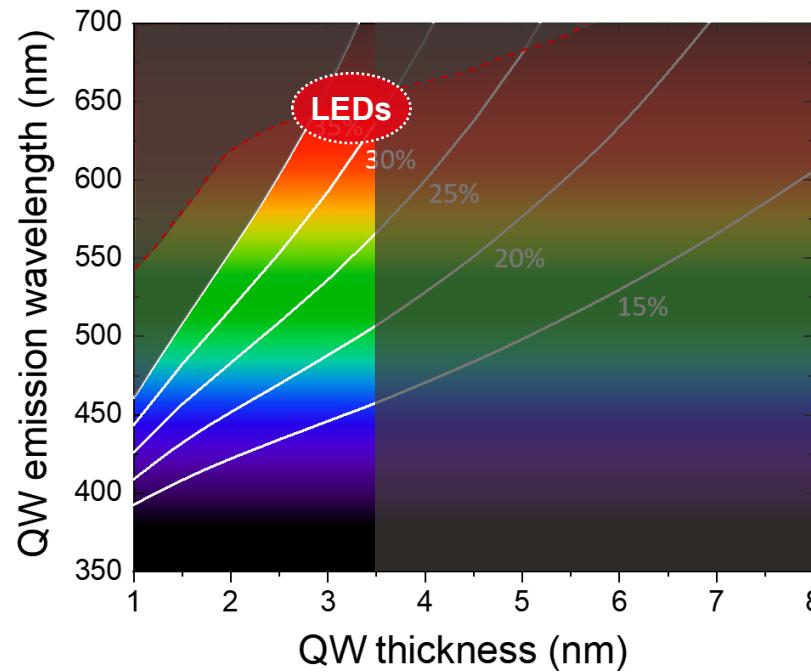
$$x = 15\% \Rightarrow F \approx 1.5 \text{ MV/cm}$$

Red emission (~610 nm) with 15% In and $L_w = 8 \text{ nm}$



How to push the wavelength to the Red ?

- InGaN/GaN QW emission wavelength



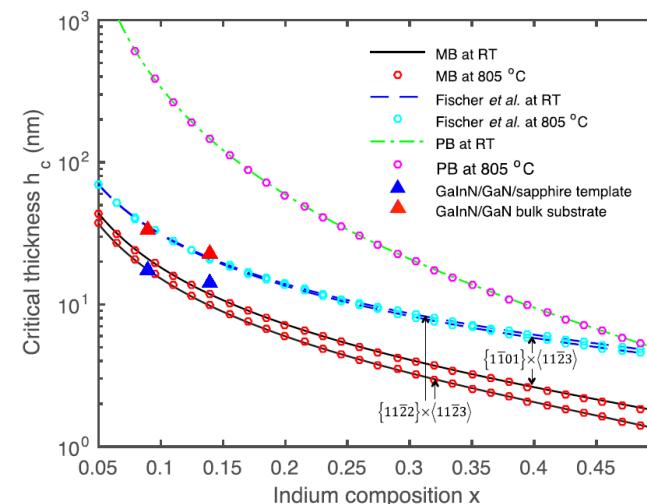
InGaN on GaN
critical thickness

G. Ju *et al.*, APL 110, 262105 (2017)
Amano's group

$$E_{QW} = E_g^{\text{InGaN}} + E_{conf}^{e,h} - qL_wF \text{ Indium content}$$

« ... less than 35% In
composition is normally
reported. ”

N. Hu *et al.*, APL 121, 082106 (2022) - Amano's group

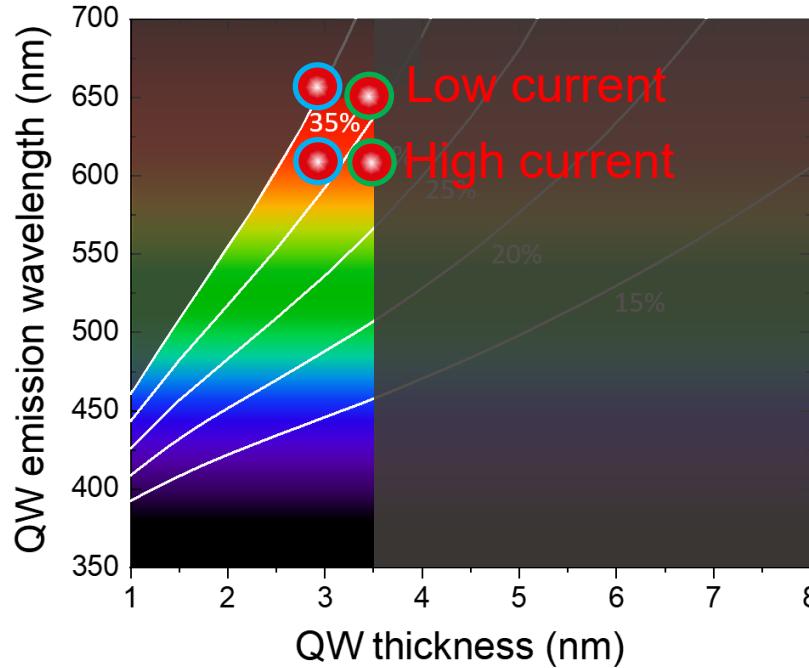


- The parameter space for red LEDs is quite limited

In content = 30-35%
 L_w = 3-3.5 nm

How to push the wavelength to the Red ?

- Red-LEDs on GaN/sapphire template



Paper 12421-61

2014 : Toshiba J.-I. Hwang *et al.*, APEX 7, 071003 (2014)

$x = 0.35, L_w = 3 \text{ nm}, \lambda_{\max} = 655 \text{ nm}$

At 20 mA and 629 nm:

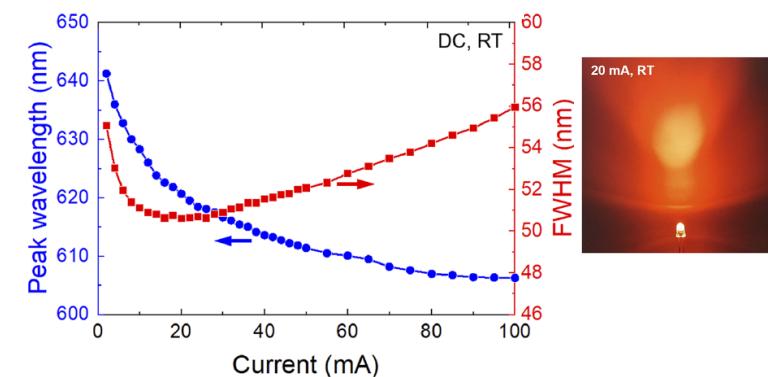
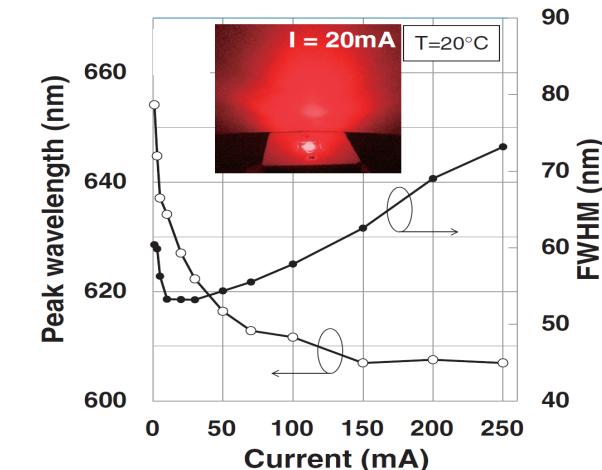
- Power = 1.1 mW
- EQE = 2.9%
- WPE = 1.3%

2022 : KAUST D. Iida *et al.*, AIP Advances 12, 065125 (2022)

$x = 0.32, L_w = 3.5 \text{ nm}, \lambda_{\max} = 641 \text{ nm}$

At 20 mA and 621 nm:

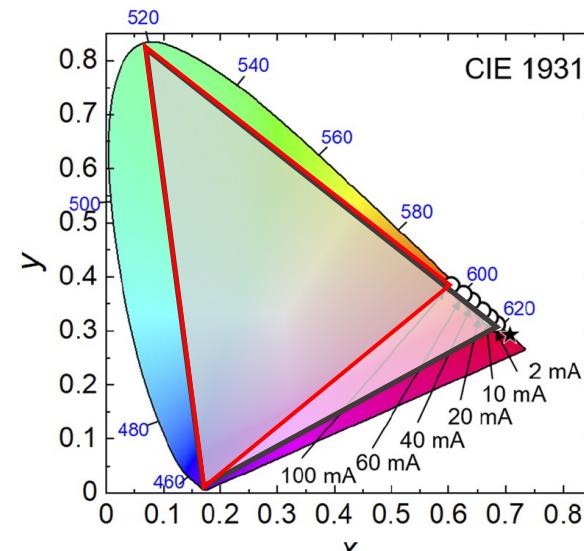
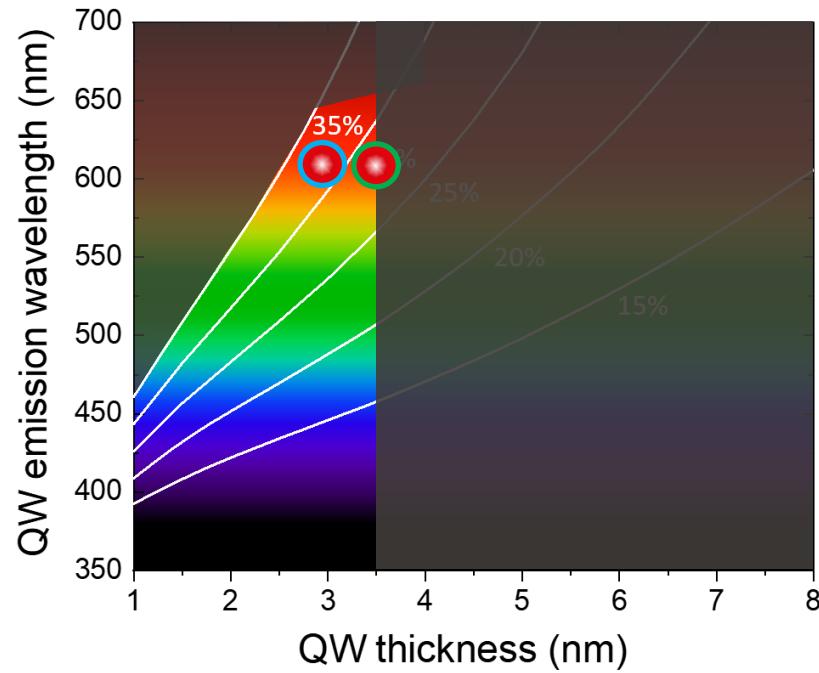
- Power = 1.7 mW
- EQE = 4.3%
- WPE = 2.9%



How to push the wavelength to the Red ?

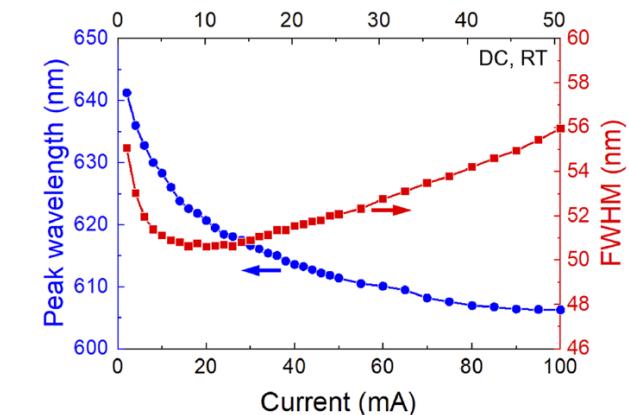
- Red-LEDs on GaN/sapphire template

At high current density, due to the field screening and band filling, the emission is barely red (610 nm)



D. Iida *et al.*, AIP Advances **12**, 065125 (2022)

The color gamut strongly depends on the “red” wavelength and the FWHM

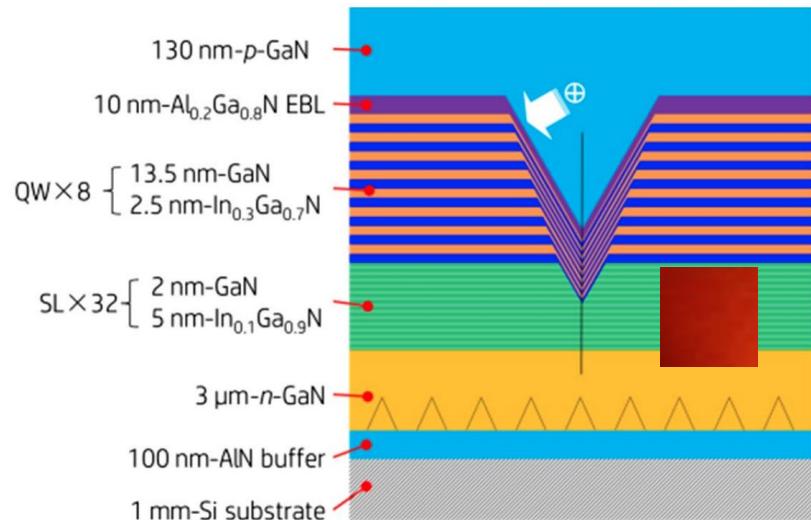


EL peak \neq dominant wavelength

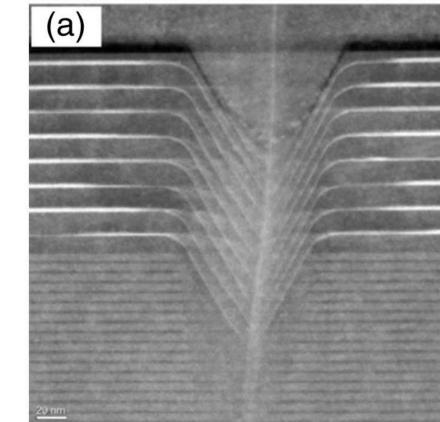
- Extend further the emission wavelength
- Reduce the emission linewidth
- Improve the efficiency

Higher In content \Rightarrow Decrease the strain

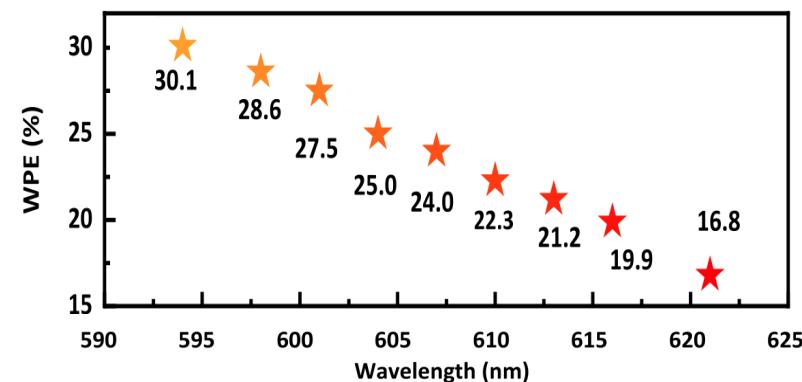
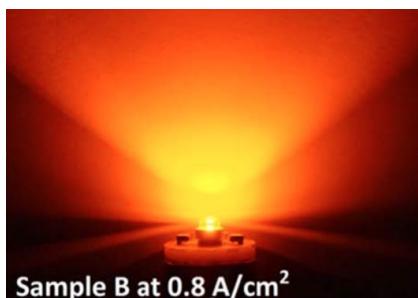
- Strain relaxation from surface morphology and growth on Si(111)



- Growth on silicon
⇒ tensile strain
- Large V-pits
⇒ strain relaxation
⇒ Lateral hole injection



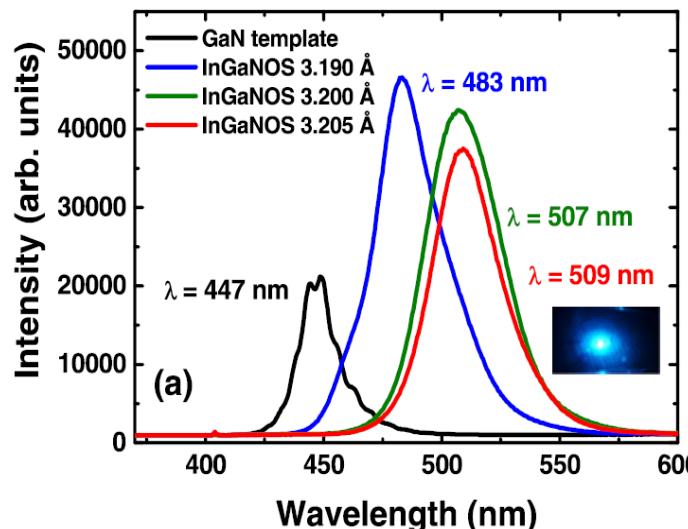
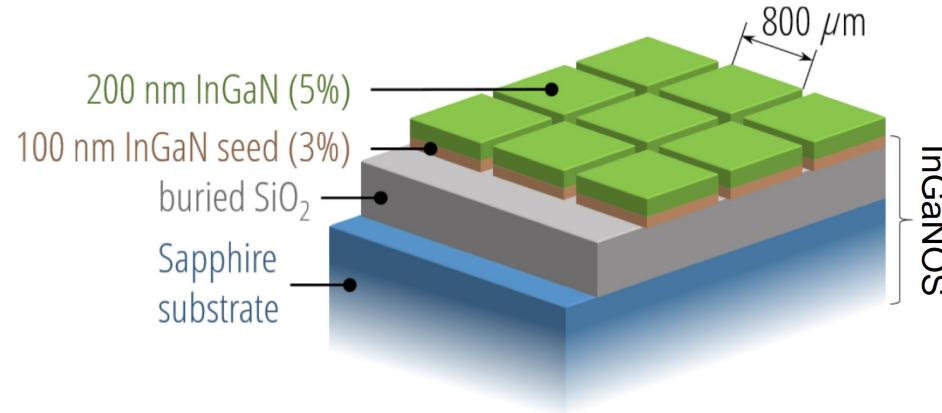
F. Jiang *et al.*, Photonics Research 7, 144 (2019) – Nanchang U.



- WPE = 16.8% at 621 nm
Large devices 1x1 mm²

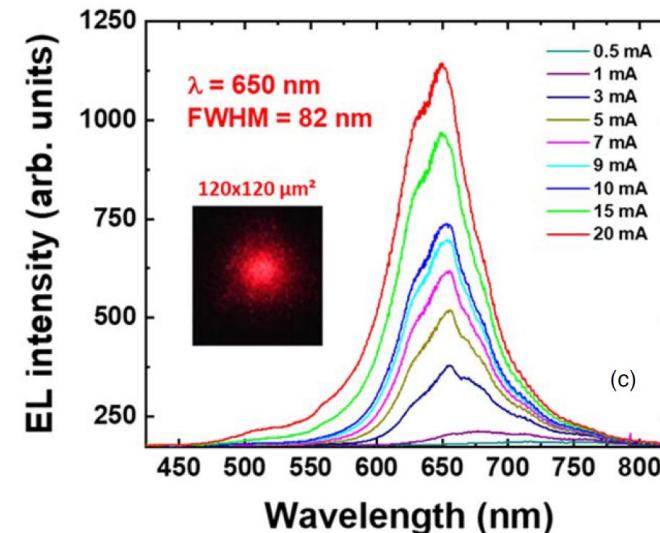
S. Zhang *et al.*, Photonics Research 8, 1671 (2020)
Nanchang U.

- Pseudo-InGaN substrate (InGaNOS)



A. Even *et al.*, APL 110, 262103 (2017)
CEA/LETI and Soitec

- 200nm-thick $\text{In}_x\text{Ga}_{1-x}\text{N}$ ($x=0.015-0.08$) layer on GaN/sapphire
- Transferred using Soitec's Smart Cut™ technology onto a compliant layer
- The InGaN layer is patterned to enable strain relaxation



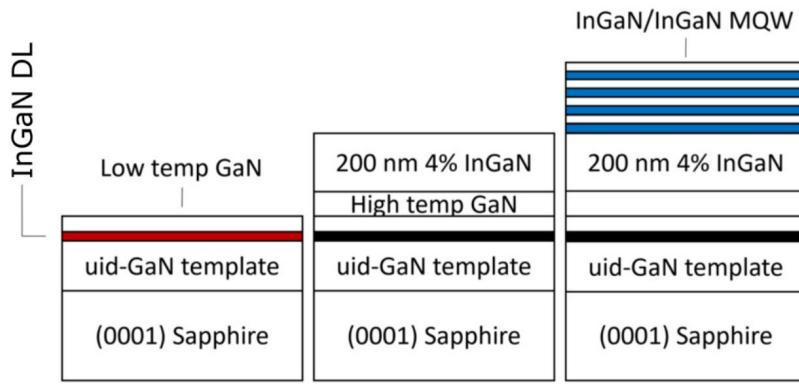
A. Dussaigne *et al.*, APEX 14, 092011 (2021)
CEA/LETI and Soitec Paper 12441-20

- In content ~40%
- 650 nm with small blueshift

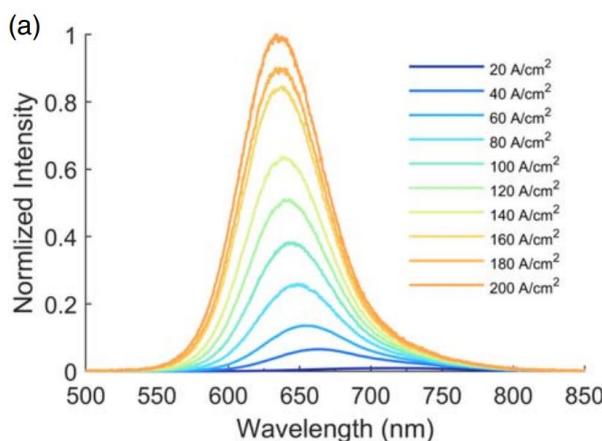
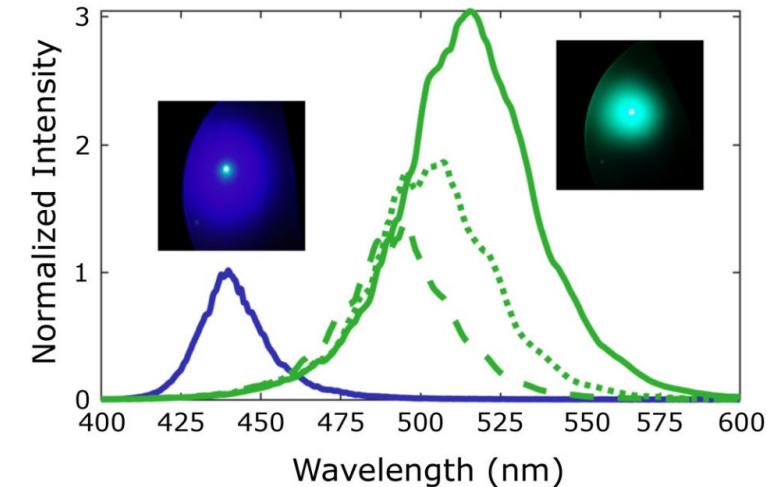
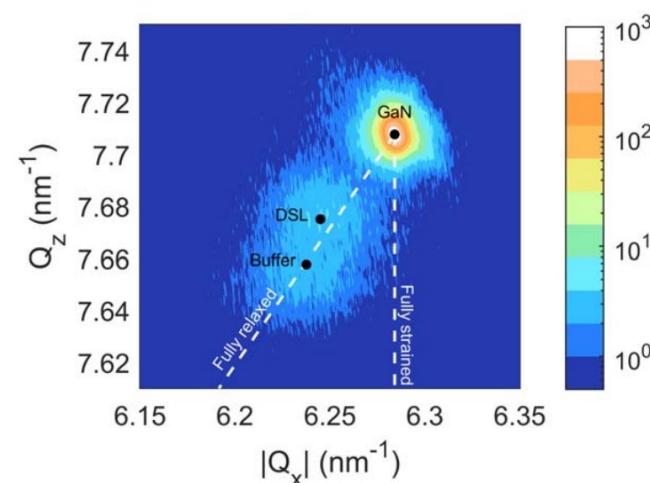
IQE = 10%
EQE = 0.14% (with a LEE<4%)

InGaNOS discontinued

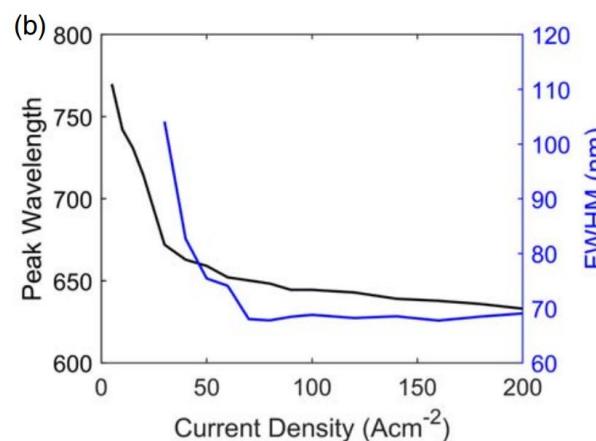
- Pseudo-InGaN substrate *via* decomposition layer



P. Chan *et al.*, APL **119**, 131106 (2021) - UCSB



P. Chan *et al.*, APEX **14**, 101002 (2021) - UCSB



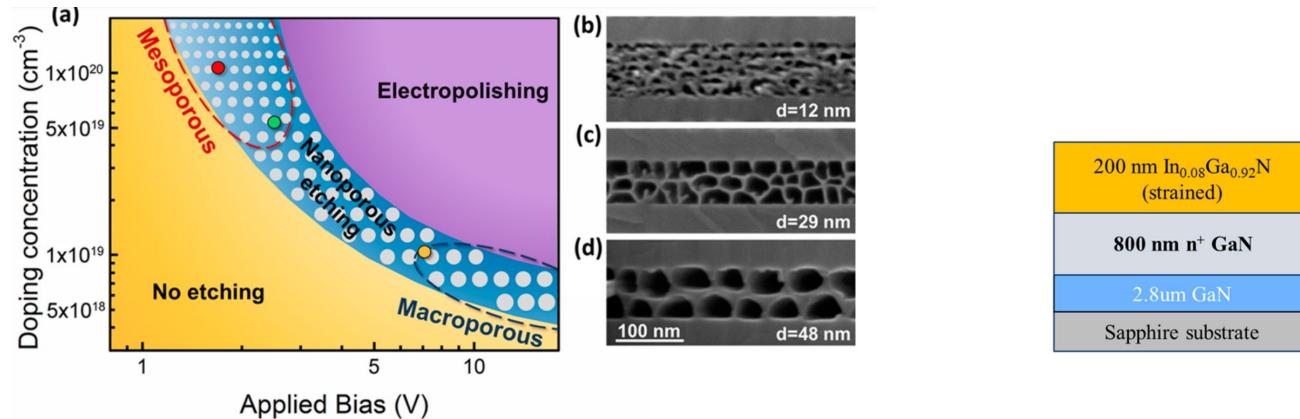
Paper 12441-57

- 770 nm at low current
- 633 nm at 200 A.cm⁻²

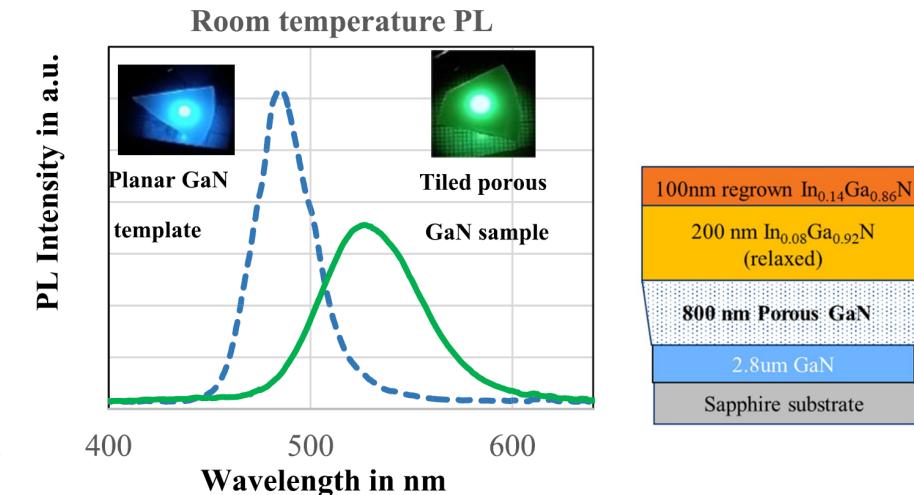
EQE = 0.05% (due to poor surface morphology)

Less strain for more indium

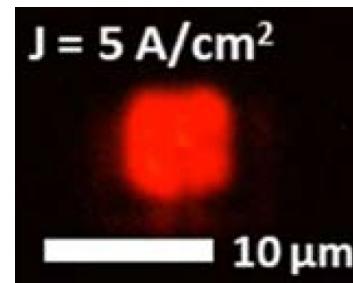
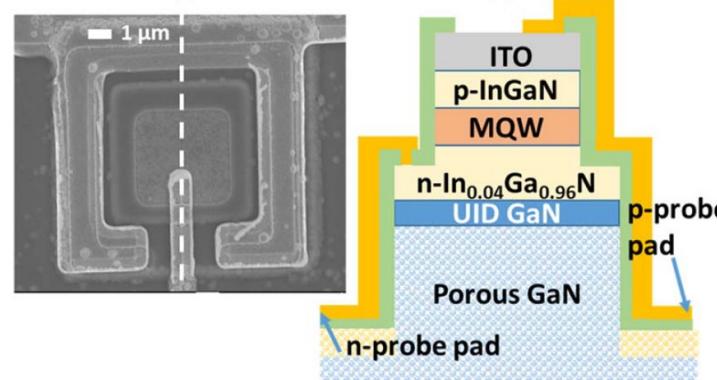
- Strain relaxation from porous GaN



C. Zhang *et al.*, ACS Photonics **2**, 980 (2015) - Yale



S.S. Pasayat *et al.*, Semicond. Sci. Technol. **34**, 115020 (2019) - UCSB



- 632 nm at 10 A.cm^{-2}
- 6 $\mu\text{m} \times 6 \mu\text{m}$ μ LEDs

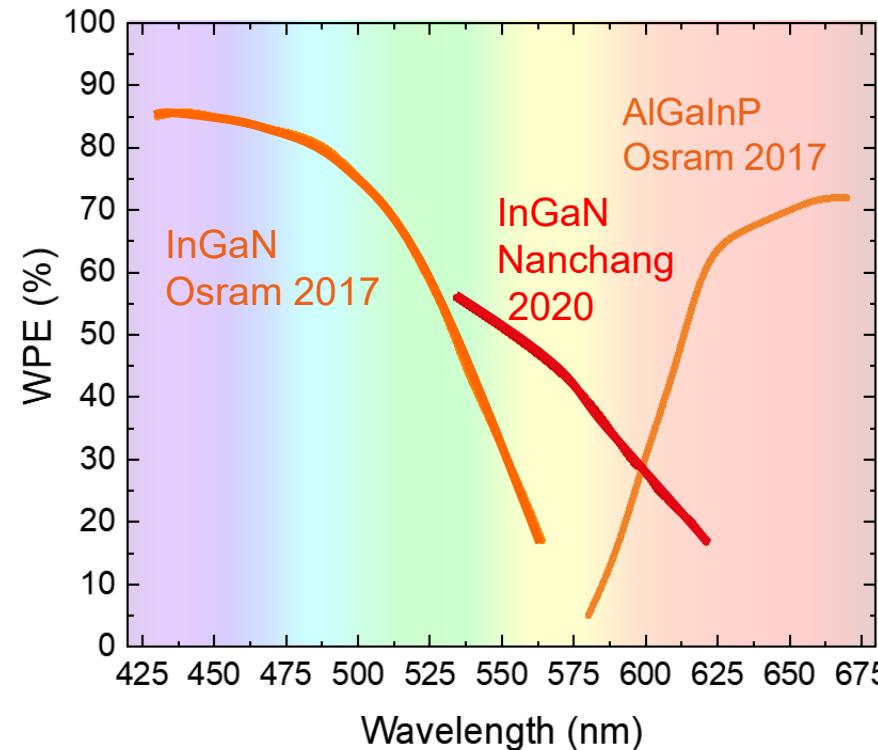
EQE = 0.2%

S.S. Pasayat *et al.*, APEX **14**, 011004 (2021) - UCSB

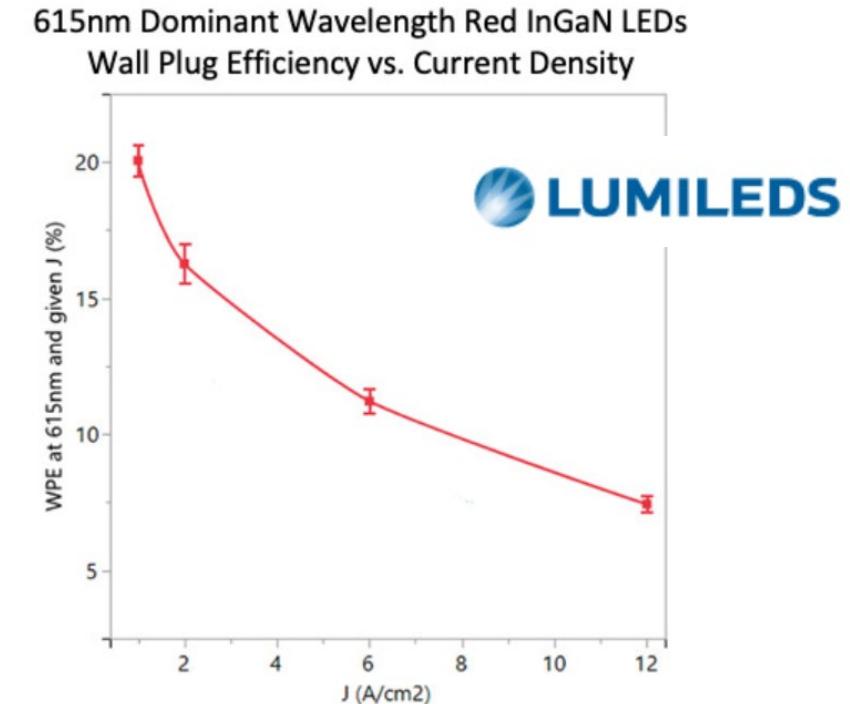
R. Oliver, Cambridge U. Paper 12441-65

Conclusion

- WPE of InGaN based LEDs – state of the art



- ➔ The green gap is closing
- ➔ The III-nitrides may compete for red microLEDs
- ➔ InGaN pseudo-substrates are needed.



<https://lumileds.com/lumileds-continues-to-lead-advances-for-red-ingan-leds/>

Electrical injection

$$n = J\tau/qd$$

A few remarks:

- The effective recombination time τ depends on the carrier density
- The Auger recombination term is significant only at high injection
- The carrier density depends on the active region thickness

Geometry of the active region:

- homojunction: $d = L_{Dn} + L_{Dp}$ (1-10 μm) (No RT operation)
- heterojunction: $d = 100 \text{ nm}$
- quantum well: $d = 1-10 \text{ nm}$

Laser diodes

Population inversion

n increases with the current

$$n = \int_{E_c}^{\infty} \rho_c(E) \frac{1}{1 + e^{\left[\frac{E - E_{Fc}}{kT} \right]}} dE \quad E_{Fc} (E_{Fv}) \uparrow$$

and the absorption is given by

$$\alpha(\omega) = -\gamma(\omega) = \alpha_0(\omega) [f_v(\hbar\omega) - f_c(\hbar\omega)] \quad \text{where } \gamma \text{ is the gain}$$

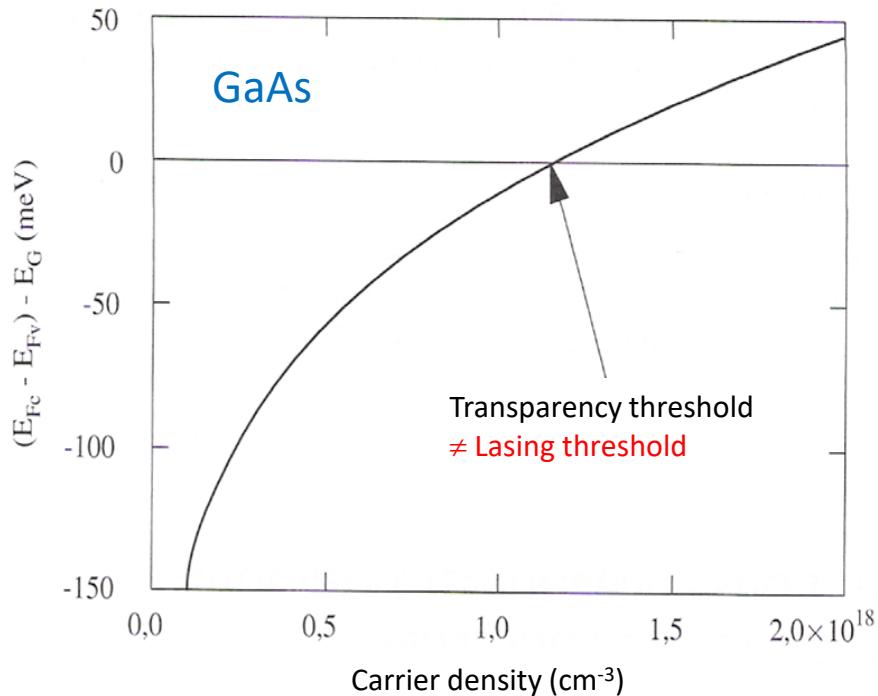
When α is negative \Rightarrow stimulated emission

$$f_c(\hbar\omega) \geq f_v(\hbar\omega) \Rightarrow E_{Fc} - E_{Fv} \geq \hbar\omega \geq E_g$$

Bernard-Duraffourg condition

Laser diodes

Transparency threshold



The material becomes transparent when

$$E_{F_c} - E_{F_v} = E_g$$

Minimum requirement to fulfill the Bernard-Duraffourg condition

$t = 1 \mu\text{m}$ $\Rightarrow J_{tr} \sim 16 \text{ kA/cm}^2$ bulk (1960)

$t = 100 \text{ nm}$ $\Rightarrow J_{tr} \sim 1.6 \text{ kA/cm}^2$ heterostructure (1970)

$t = 10 \text{ nm}$ $\Rightarrow J_{tr} \sim 160 \text{ A/cm}^2$ quantum well (1980)

Laser threshold

$$\text{gain} = \text{losses} \Rightarrow \Gamma \gamma_{thr}(h\nu) = \alpha_p + 1/(2L) \times \ln(1/R_1 R_2)$$

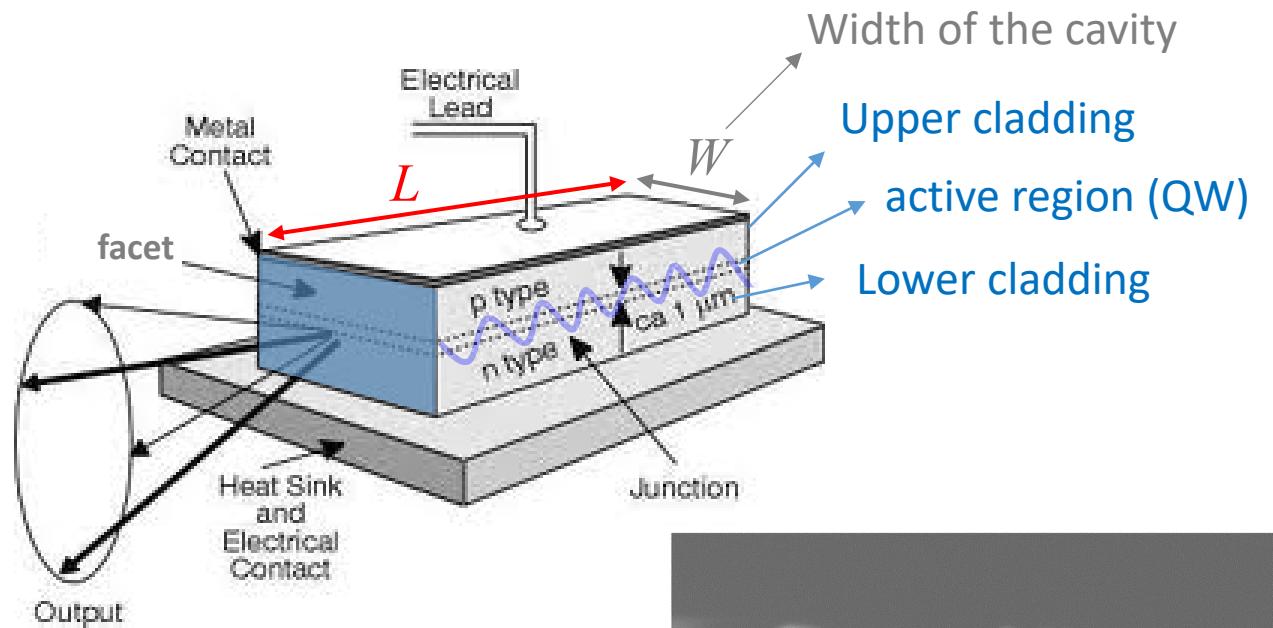
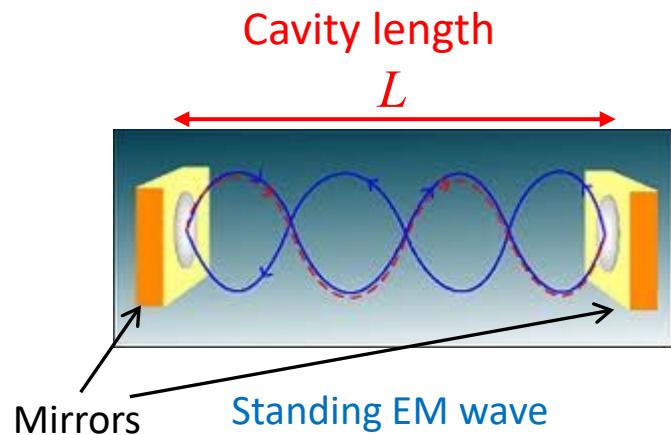
Modal gain

Γ is the confinement factor

Edge-emitting laser diode

Laser cavity

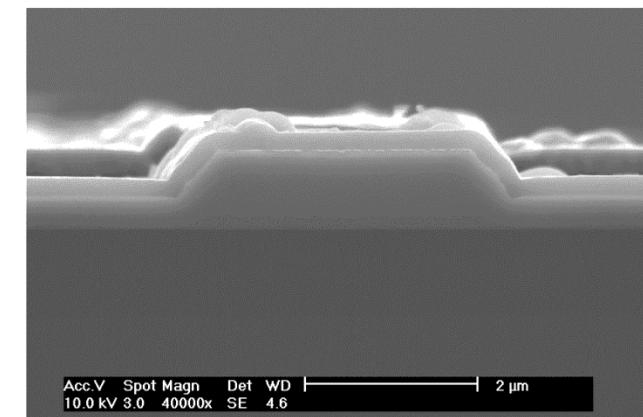
Resonant cavity \Rightarrow optical feedback



The mirrors are achieved by crystal cleavage. This defines atomically flat planes.

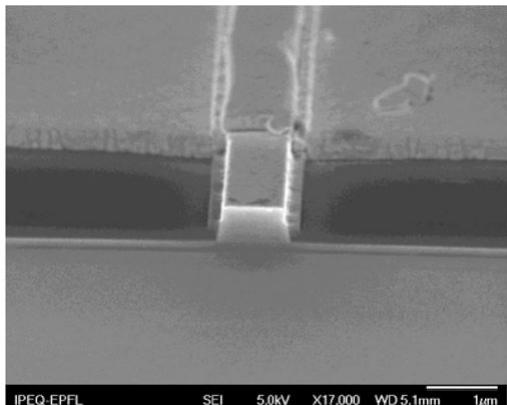
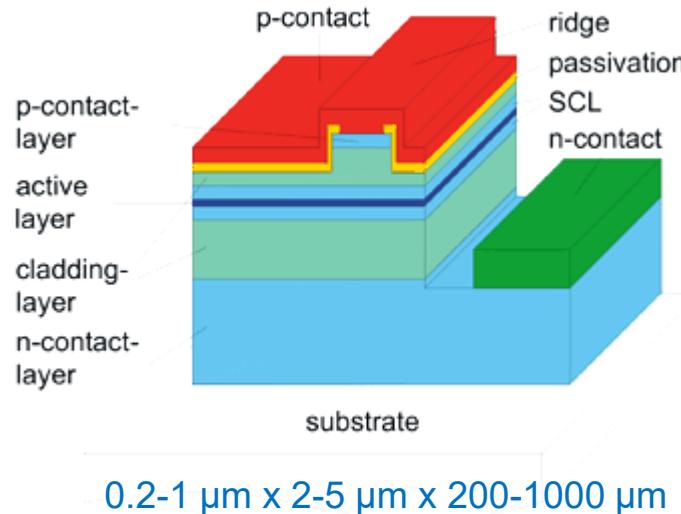
The reflectivity is given by $R_m = (n_{sc} - 1)^2 / (n_{sc} + 1)^2$ e.g., for GaAs, $R_m = 0.32$

The facet reflectivity is further increased by dielectric surface coating (DBRs).

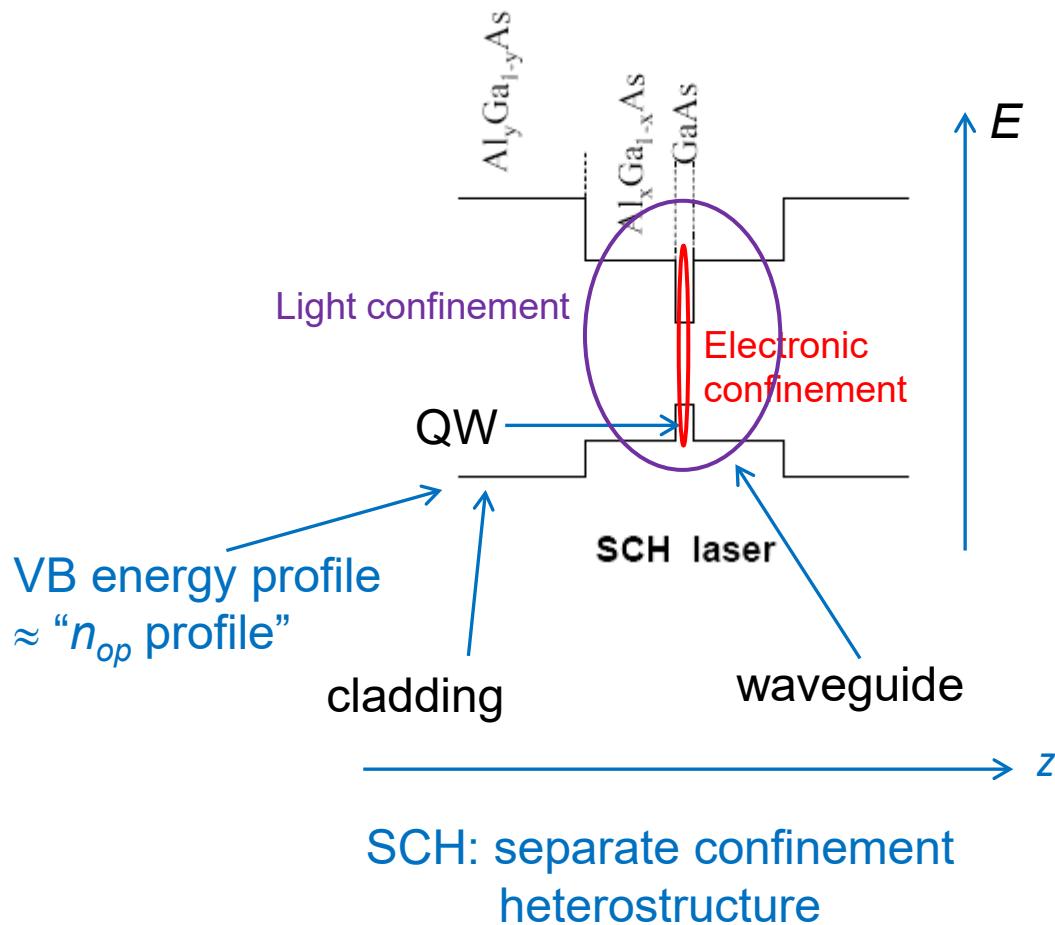


Edge-emitting laser diode

Laser diode structure

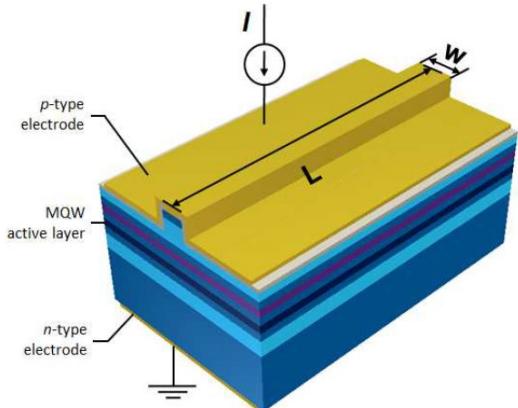


“LED” + cavity (mirrors)

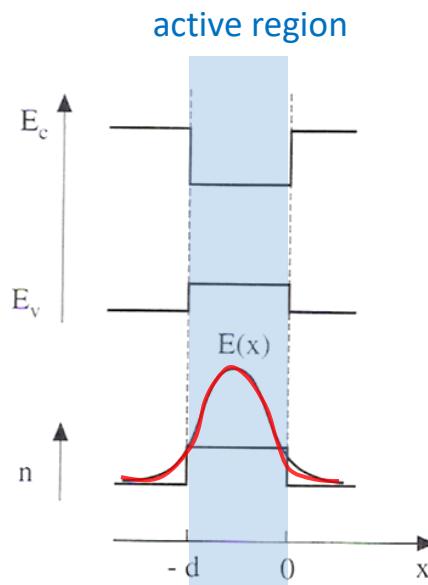
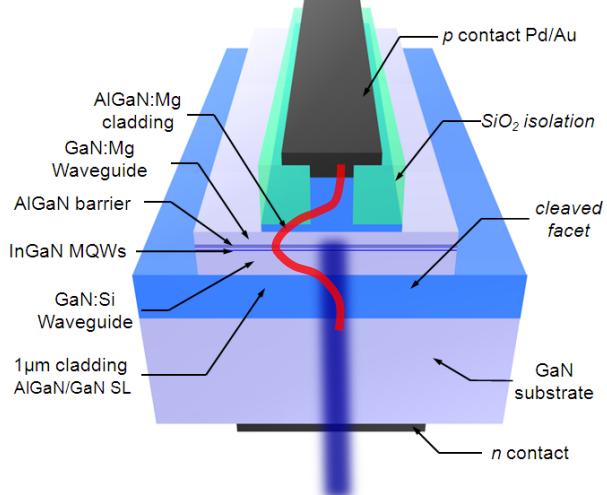


Edge-emitting laser diode

Confinement factor and modal gain



Optical waveguide:



The confinement factor (Γ) is given by the overlap between the optical mode and the active region (e.g. QWs):

$$\Gamma = \frac{\int_{-d}^0 |E(x)|^2 dx}{\int_{-\infty}^{+\infty} |E(x)|^2 dx}$$

For a QW-based LD (SCH), $\Gamma = 0.5\text{-}5\%$

For a double heterostructure (DHS), $\Gamma \sim 1$ ($d \approx 100$ nm)

Then, the modal gain is

$$\gamma_m = \Gamma \gamma$$

Edge-emitting laser diode

Laser threshold

In an ideal case, i.e. without any losses, the laser threshold would be the transparency threshold ($n_{tr} = n_{thr}$)

However, in real devices, parasitic absorption is always present ($n_{tr} < n_{thr}$)

The laser threshold is then defined as **Gain = Losses**

$$\gamma_m = \Gamma \gamma$$

γ increases with injected current

- Internal losses (parasitic light absorption, eg. not the active region)
- Mirror losses

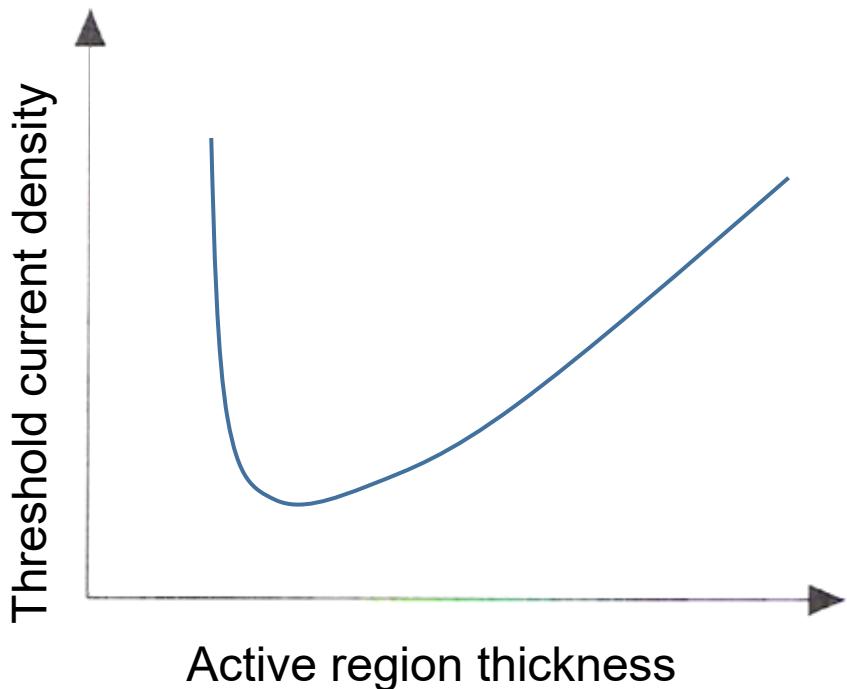
Condition for lasing: Modal gain = losses $\Rightarrow \Gamma \gamma_{thr} = \alpha_p + 1/(2L) \times \ln(1/R_1 R_2)$

Modal gain Parasitic losses

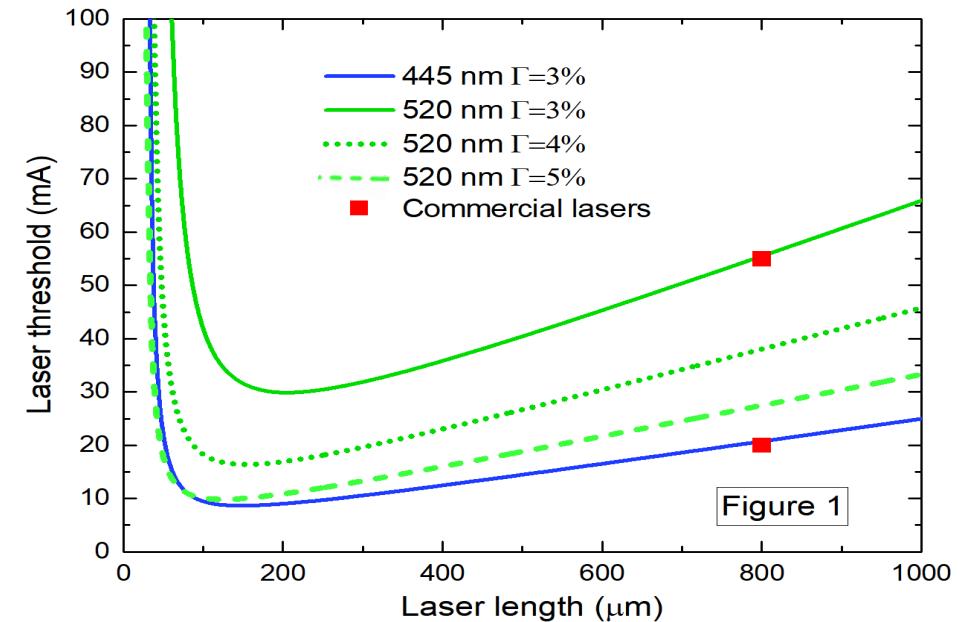
R_1, R_2 are the mirror reflectivity

Edge-emitting laser diode

Laser threshold



For a given laser diode size, there is a minimum threshold current density which is determined by the tradeoff between carrier density in the active region and the confinement factor.



Evolution of the laser threshold as a function of cavity length, confinement factors, and different emission wavelengths

Edge-emitting laser diode

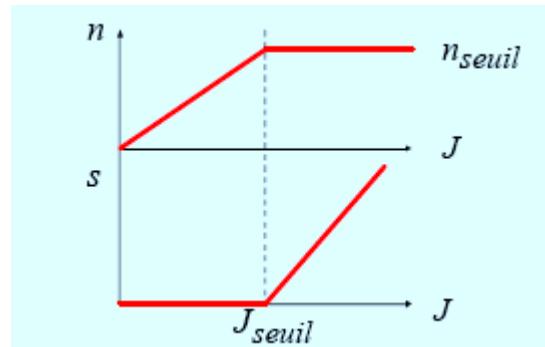
Light output characteristics

1. Below the transparency threshold: linear dependence with the current
2. Above the transparency threshold: amplified spontaneous emission (ASE),
 \Rightarrow superluminescence $n_{tr} < n < n_{thr}$
3. When $\Gamma\gamma$ (modal gain) equals the losses: laser oscillations start (strong linewidth reduction)

Important: once the lasing threshold is reached, the carrier density is clamped (n is constant)



each newly added electron gives rise to 1 stimulated photon (x IQE)

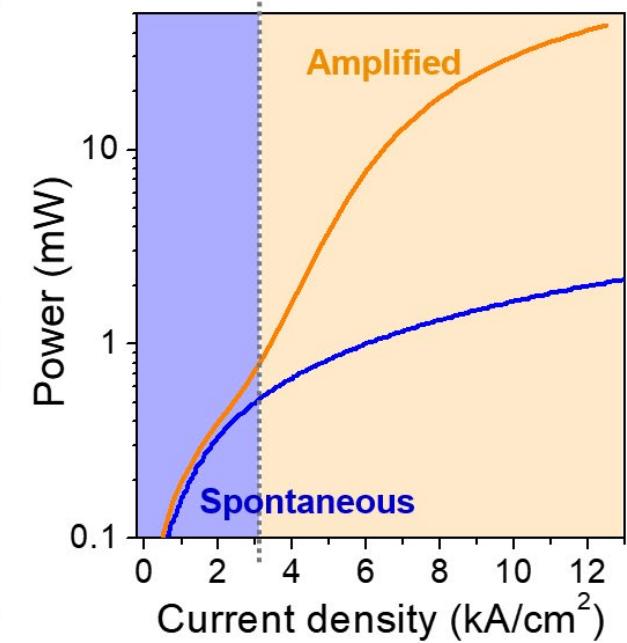
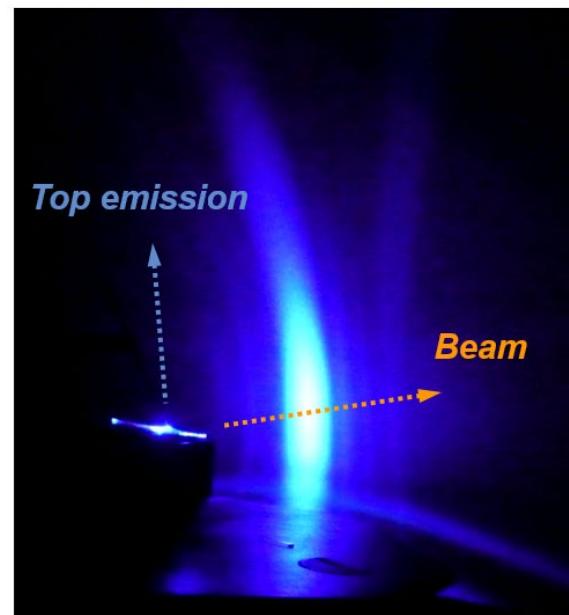
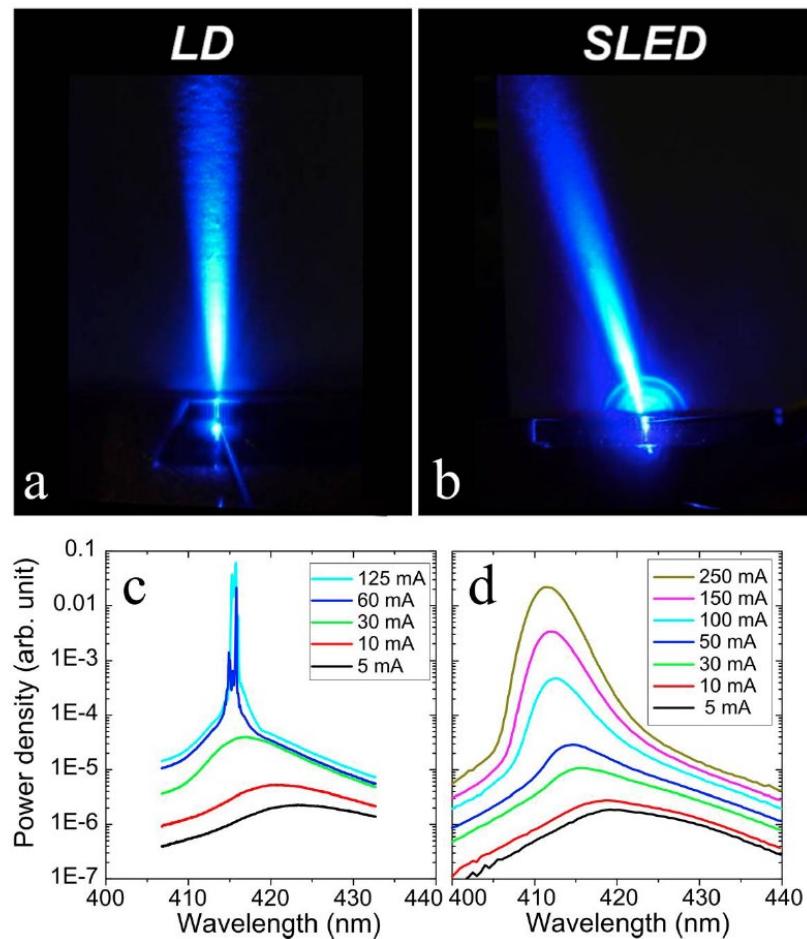


s is the number of stimulated photons in the cavity

Edge-emitting laser diode

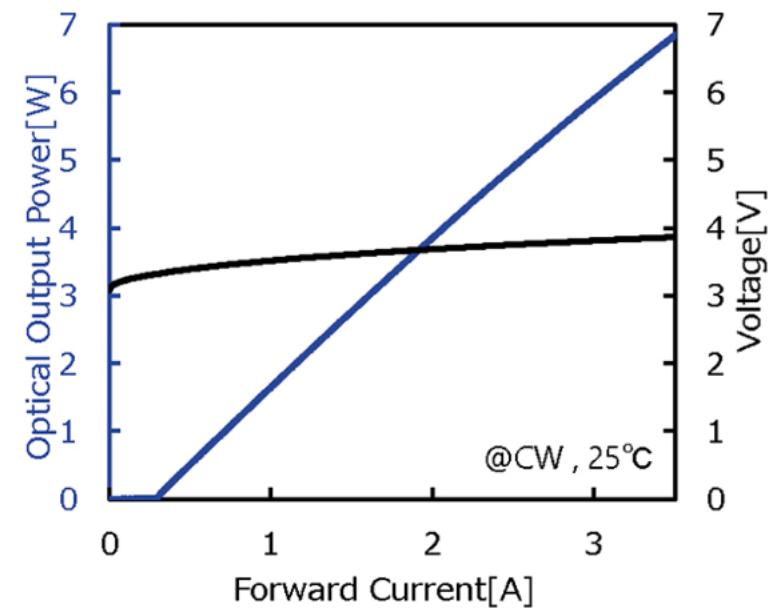
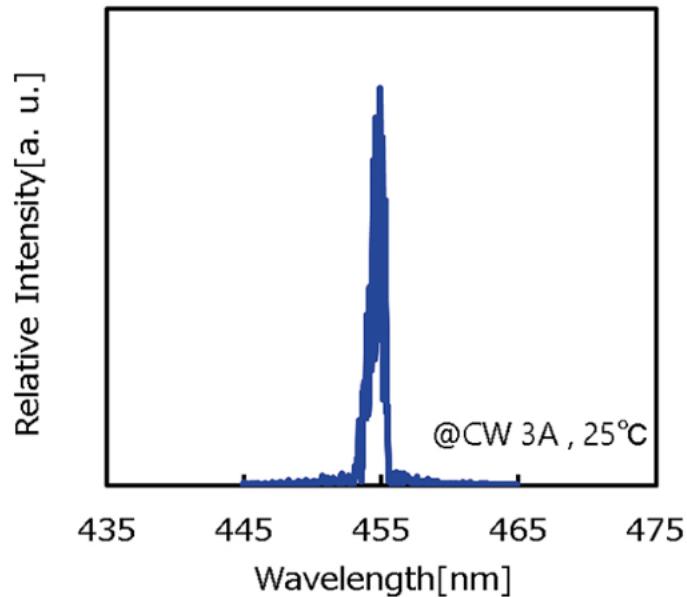
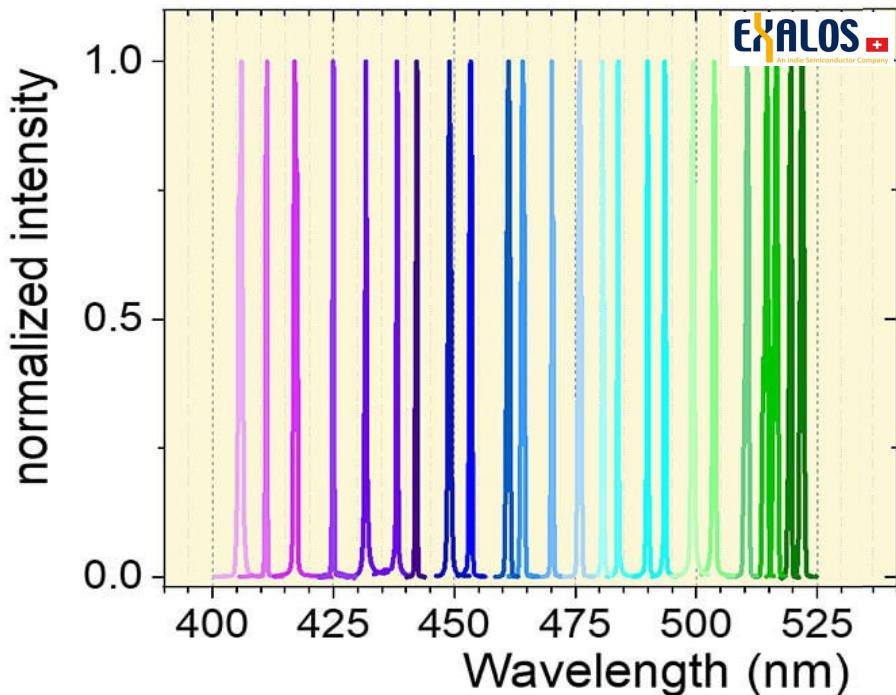
Lasing and amplified spontaneous emission

Laser diode versus superluminescent LED



Note: a superluminescent LED (SLED) is a laser diode structure without any internal light reflection (no cavity effect=no feedback)

Edge-emitting laser diode

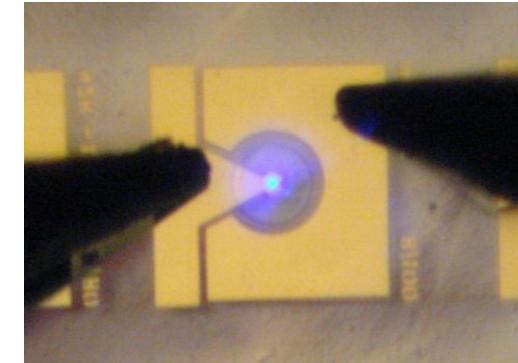
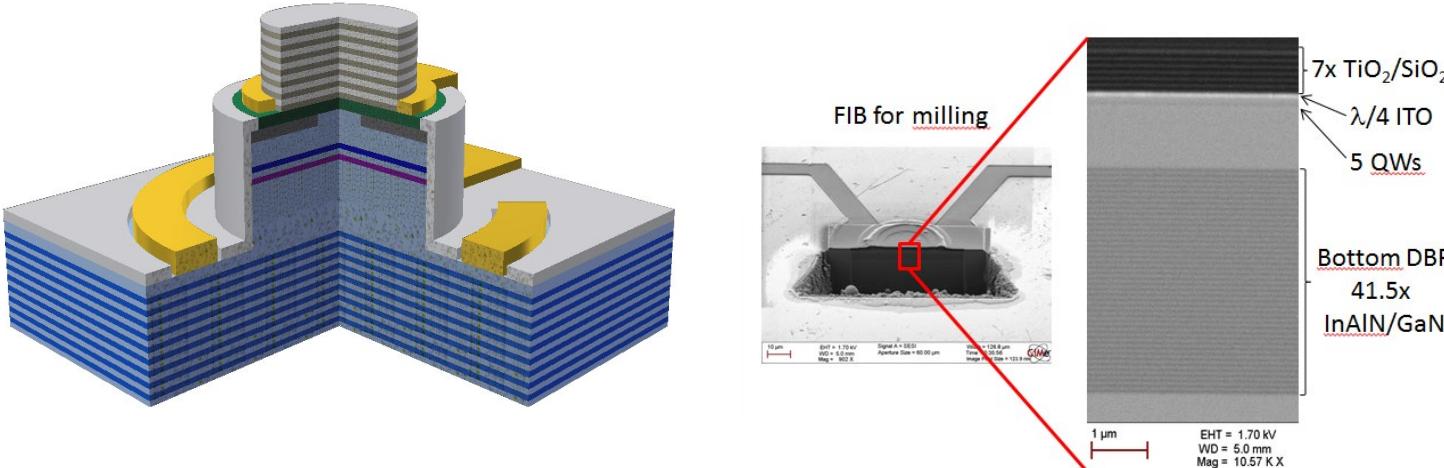


[https://compoundsemiconductor.net/article/117364/Building better blue and green lasers](https://compoundsemiconductor.net/article/117364/Building_better_blue_and_green_lasers)

WPE > 50% at 450 nm
WPE > 25% at 525 nm

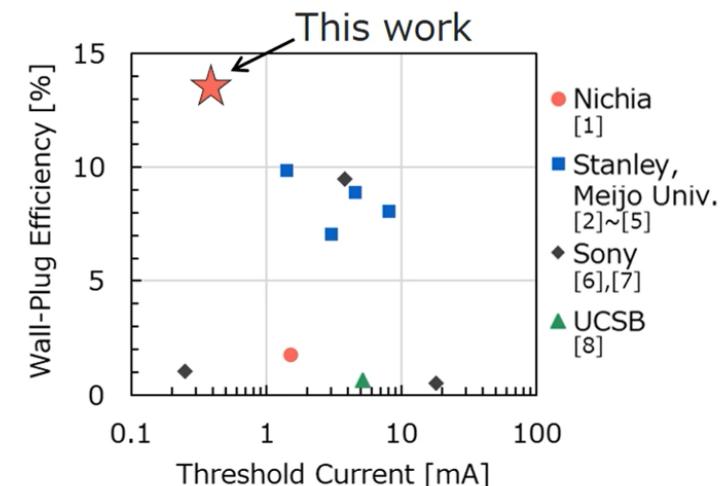
VCSELs

Vertical cavity surface emitting laser (VCSEL)



Appl. Phys. Lett. 101, 151113 (2012)

■ WPE vs I_{th} of blue VCSELs



5 March 2021

Blue and green GaN-based vertical-cavity surface-emitting lasers with AlInN/GaN DBR

Kenichi Terao, Hitoshi Nagai, Daisuke Morita, Shingo Masui, Tomoya Yanamoto, Shin-ichi Nagahama

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